

# Study of the Structural and Elastic Properties of $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$ Material

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**Abstract-**  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material was synthesized by sol-gel combustion method using nitrates and citric acid. The structure and elastic properties of the synthesized material were investigated using powder x-ray diffraction (XRD) and Fourier transform infrared (FTIR) spectroscopy. The structural parameters including lattice constant, unit cell volume, and crystallite size of the synthesized material were estimated. The bulk modulus, Young's modulus, rigidity modulus, longitudinal wave velocity, transverse wave velocity, and mean velocity were also evaluated from XRD and FTIR analysis values. XRD result revealed that the  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material possessed a single-phase cubic spinel structure with Fd-3m space group without the presence of other phase impurities. The lattice parameter and the average crystal size of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material were also found to be 0.8538 and 58.3 nm, respectively. The FT-IR analysis revealed the presence of two strong absorption bands of the tetrahedral and octahedral groups in the  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material. The combined results of XRD and FTIR confirmed that the  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material is mechanically stable.

**Keywords:** Sol-gel combustion, Structure, Elastic property, Structural parameter, cubic spinel.

## 1. INTRODUCTION

Ferrites are ferrimagnetic oxides consisting of ferric oxide and metal oxides. Ferrites with spinel structure belong to the important class of magnetic materials. The combination of the structural and mechanical properties makes ferrite useful in many technological applications [1]. Spinel ferrites have the general formula of  $\text{MFe}_2\text{O}_4$ , where, M usually represents one or more than one of the divalent transition metals such as Mn, Fe, Co, Ni, Cu, Zn, or other metals [2]. Other combinations of equivalent valences are also possible, such as substitution of some or all of the trivalent iron in a ferrite compound with other trivalent metal ions, such as  $\text{Al}^{3+}$ ,  $\text{Cr}^{3+}$ , and  $\text{Mn}^{3+}$  ions. A unit cell of spinel ferrite material contains 32 oxygen atoms in a cubic close packing with 8 tetrahedral and 16 octahedral occupied sites [3]. Among the spinel ferrites, the inverse type is particularly interesting due to its high magneto crystalline anisotropy and unique magnetic structure [3].

$\text{NiFe}_2\text{O}_4$  is an inverse spinel ferrite in which the ferric ions are equally distributed in the tetrahedral sites and octahedral sites [4]. Some of  $\text{NiFe}_2\text{O}_4$  applications are including high density magnetic recording media, magnetic refrigeration, magnetic liquids, microwave absorber and repulsive suspension for levitated railway systems [5,6].  $\text{NiFe}_2\text{O}_4$  exhibits unusual physical and chemical properties when its size is reduced in to the nanolevel. This ferrite material is prepared today by various techniques in different form and intensive research work is conducted to obtain better performing  $\text{NiFe}_2\text{O}_4$  based spinel ferrite materials.

Nowadays, the synthesis of spinel ferrite nanoparticles has been intensively studied, because of their remarkable electrical and magnetic properties and wide practical application to information storage system, ferrofluid technology, magneto caloric refrigeration, catalysis, and medical diagnostics [2]. These properties are dependent on chemical composition and microstructural characteristics, which can be controlled in the fabrication and synthesis processes [2]. The principal role of the preparation conditions on the optical, electrical and microstructure features of the ferrites have been discussed in several papers; various methods of synthesizing spinel nickel ferrite nanoparticles have been reported [7]. These methods include sol-gel, hydrothermal, solid state reaction, chemical co-precipitation, molten salt etc. [7]. Among these, sol-gel combustion method is widely used for the synthesis of spinel ferrite. In this study, sol-gel combustion method was chosen for the preparation of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material. The structure and elastic properties of this material were investigated by using x-ray diffraction (XRD) and Fourier transform infrared (FTIR) spectroscopy characterization techniques.

## 2. MATERIALS AND METHODS

### 2.1. Synthesis Procedures

$\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material was synthesized by sol-gel combustion method.  $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Mn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  and citric acid monohydrate ( $\text{C}_6\text{H}_8\text{O}_7 \cdot \text{H}_2\text{O}$ ) were used as raw materials for the synthesis process. The stoichiometric amounts of  $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Mn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ,  $\text{Fe}(\text{NO}_3)_3$

.9H<sub>2</sub>O and citric acid monohydrate (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>.H<sub>2</sub>O) precursors was weighed out and dissolved in double distilled water and stirred for about 20 minutes. The stoichiometric amount of citric acid was dissolved in distilled water in separate beaker and stirred for about 20 minutes. The citric acid solution was further added in to the solution of nitrates. This solution was further stirred for about 15 minutes with magnetic stirrer and ammonia solution was then added under constant stirring in order to make the pH value 7. The resulting solution was continuously heated on the magnetic stirrer hotplate at 60°C to form a gel. When the obtained gel is further heated at 90°C, combustion process was conducted. The obtained black powder was ground in an agate mortar for about 2 hours and heated at 950°C for 10 hours. Finally, this powder was ground for about using agate mortar and pestle.

## 2.2. Material Characterizations

X-ray powder diffraction (XRD) is one of the most valuable techniques for analyzing the following crystallographic properties: crystallite size, lattice parameter, purity, etc. of materials in powder form. It is also important to point out the unknown compounds by comparing diffraction data against a database of known powder diffraction patterns maintained by joint Committee on Powder Diffraction Standards (JCPDS). In this study, the x-ray powder diffraction pattern of Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material was recorded using X-ray diffractometer (XRD-7000S diffractometer) with a Cu K $\alpha$  radiation of wavelength  $\lambda = 1.5406 \text{ \AA}$  source. The measurement was conducted between a diffraction angle of  $2\theta = 10^\circ$  and  $80^\circ$ .

Fourier transform infrared (FTIR) spectroscopy is a non-destructive characterization technique that utilizes for the identification of types of chemical bonds in the compounds by producing infrared absorption spectra. It is also used for identification of the structure of ceramic materials from the frequencies of the vibration modes. In this work, FT-IR spectroscopy measurement were accomplished using transmittance method using ALFA-T instrument with potassium Bromide (KBr) as IR window in the wave number region between 350 and 4000 cm<sup>-1</sup>.

## 3. RESULTS AND DISCUSSION

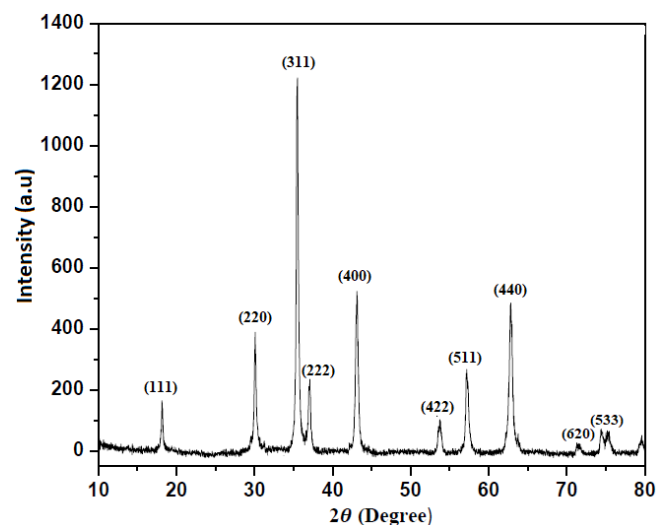
### 3.1. XRD analysis

X-ray powder diffraction is a technique for analyzing the crystallite size, lattice parameter, phase identification and the unit cell volume of materials. The room temperature XRD pattern of Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material prepared by using sol-gel combustion method calcined at a

The lattice constants (a) and the unit cell volume of Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material were calculated by the least-

temperature of 950°C for 10 hours is shown in Fig. 1. As it is observed from the figure, broadened, sharp and well defined XRD peaks are observed in Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material. This indicates that Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material possess a good crystalline structure. Most importantly, The broadening of the peaks indicates the nanocrystalline nature of synthesized material [8]. This is due to the fact that in nano sized particles there are insufficient diffraction centers, which causes the diffraction lines brooding. The XRD pattern also reveals the pure phase formation of Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material without any impurity. The crystalline phase of the synthesized material is identified as a single phase cubic spinel structure with Fd-3m space group of NiFe<sub>2</sub>O<sub>4</sub> material. Moreover, the obtained XRD peaks are also indexed using the Joint Committee on Powder Diffraction Standards (JCPDS) card with good agreement for NiFe<sub>2</sub>O<sub>4</sub> (card no. 86-2267).

As shown in the Figure, the diffraction peaks observed at  $2\theta$  values of 30.1, 35.45, 36.87, 43.08, 53.39, 56.12, 62.48, 65.22 and 74.20° correspond to (111), (220), (311), (222), (400), (422), (511), (440), (620) and (533) planes, respectively. The observed peaks also confirms the formation of a spinel cubic structured in Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material synthesized by sol-gel combustion method at a temperature of 950°C for 10 hours using Ni(NO<sub>3</sub>)<sub>2</sub>.6H<sub>2</sub>O, Co(NO<sub>3</sub>)<sub>2</sub>.6H<sub>2</sub>O, Mn(NO<sub>3</sub>)<sub>2</sub>.6H<sub>2</sub>O, Fe(NO<sub>3</sub>)<sub>3</sub>.9H<sub>2</sub>O, and C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>.H<sub>2</sub>O raw materials. The observed XRD peaks have a very good agreement with the reported results by Yattainahalli *et al.* [9]. It is also reported that the Bragg planes of (422) and (440) correspond to tetrahedral and octahedral sites, respectively. In this study, the XRD peaks of the planes of (422) and (440) are also identified, indicating the presence of the tetrahedral and octahedral sites in Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material.



**Figure-1:** XRD pattern of Ni<sub>0.8</sub>Co<sub>0.15</sub>Mn<sub>0.05</sub>Fe<sub>2</sub>O<sub>4</sub> material. square fitting method from the d-spacing (d) and the Miller indices (hkl) values using equations [10];

$$a = d\sqrt{h^2 + k^2 + l^2} \text{ and}$$

$$V = a^3$$

The average crystal sizes (D) of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material was also estimated by using Debye-Scherrer formula [10];

$$D = \frac{0.9\lambda}{\beta \cos \theta}$$

were,  $\lambda$  is the wavelength of the X-ray,  $\beta$  is the full width at half maximum (FWHM) of the diffraction peak and  $\theta$  is the Bragg's angle.

The lattice parameter is found to be 0.8344 nm for  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material. Its unit cell volume is also found to be  $581 \text{ (\AA)}^3$ . This shows that the obtained results are slightly larger than the reported values ( $8.3422 \text{ \AA}$  and  $581 \text{ (\AA)}^3$ ) for the parent  $\text{NiFe}_2\text{O}_4$  compound [11]. The variation in the unit cell volume and lattice constant can be explained on the basis of Vegard's law, which states that the variation in these structural parameters are related to the ionic radii of the substituted ions. In the present study, some amount of  $\text{Ni}^{2+}$  ion (ionic radius,  $0.69 \text{ \AA}$ ) [12] is replaced by  $\text{Co}^{2+}$  ion (ionic radius,  $0.74 \text{ \AA}$ ) [13] and  $\text{Mn}^{2+}$  ion (ionic radius,  $0.83 \text{ \AA}$ ) [13]. Thus, the substitution of  $\text{Ni}^{2+}$  ion with  $\text{Co}^{2+}$  and  $\text{Mn}^{2+}$  ions results in a slight increase in lattice constant as well as unit cell volume of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material. Similar report has been made by Babu and Tatarchuk [14]. The crystal size is found to be 61.6 nm  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material, which reveals that the synthesized material has the nanocrystalline nature of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material.

### 3.2. FTIR spectroscopic study

Fourier transform infrared spectroscopy is a common technique which is used to detect the metal-oxygen ions stretching and bending vibration in compounds. Fig. 2 shows the FTIR spectra of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material synthesized by using Sol-gel combustion method at a temperature of  $950^\circ\text{C}$  for 10 hours. Different researchers reported that the metal cations in  $\text{NiFe}_2\text{O}_4$  material are located at tetrahedral and octahedral sites according to the geometric configuration of the oxygen ions nearest neighbors. Moreover, two main broad metal-oxygen absorption bands are found in the FTIR spectrum of  $\text{NiFe}_2\text{O}_4$  material. The high frequency band  $\nu_1$  observed at around  $600 \text{ cm}^{-1}$  represent the tetrahedral metal-oxygen vibration while the low-frequency band  $\nu_2$  observed at around  $400 \text{ cm}^{-1}$  represent the octahedral metal-oxygen vibration [15]. The variation in the position of the absorption bands is due to the difference in the bond strength between the metal-oxygen coordination in the octahedral and tetrahedral sites. For instance, Srivastava et al. [10] have investigated the FTIR spectrum of  $\text{NiFe}_2\text{O}_4$  material, and they reported that the absorption band

$\nu_1$  around  $597.94 \text{ cm}^{-1}$  corresponds to the stretching vibrations of iron-oxygen ions band in tetrahedral sites and  $\nu_2$  around  $397.21 \text{ cm}^{-1}$  is related to the stretching vibrations of iron-oxygen and nickel-oxygen bands in tetrahedral sites.

The room temperature FTIR spectrum of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material prepared by sol-gel combustion technique is shown in Fig. 2. As shown in the figure, the FT-IR analysis of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material confirms the presence of two strong absorption bands  $\nu_1$  and  $\nu_2$  which lie in the expected range of cubic spinel-type  $\text{NiFe}_2\text{O}_4$  material [15]. The higher frequency band  $\nu_1$  which appears at  $587 \text{ cm}^{-1}$  is assigned to the stretching vibration of the metal-oxygen bonding force of tetrahedral group in  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  lattice structure [16]. While, the lower frequency band  $\nu_2$  which appears at  $426.5 \text{ cm}^{-1}$  may represent the metal-oxygen stretching vibration of octahedral group [16]. The presence of the two bands from the IR spectrum of  $\text{Ni}_{0.8}\text{Mg}_{0.2}\text{Al}_{0.05}\text{Fe}_{1.95}\text{O}_4$  reveals the formation cubic spinel nano material. Different researchers have identified that the normal mode of vibration of the tetrahedral cluster is higher than that of the octahedral cluster. This change is related to the shorter bond length of the tetrahedral cluster and the longer bond length of the octahedral cluster [12]. As compared with the bands of  $\text{NiFe}_2\text{O}_4$  compound reported by Srivastava et al. [10], the absorption peak of in the  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  lattice is shifted towards lower wavenumber region, which is in good agreement with the lower lattice parameter, unit cell volume and crystal size calculated from the XRD pattern.

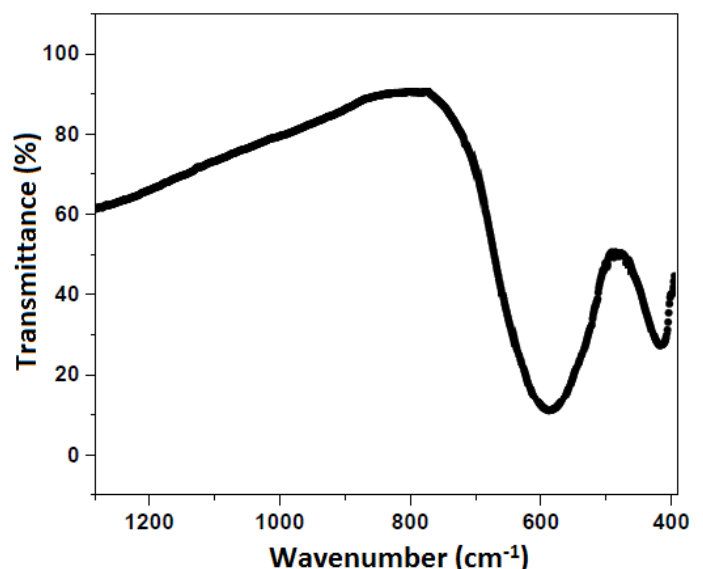


Figure-2: FTIR spectrum of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material.

The value of the force constants for tetrahedral sites  $K_T$  and octahedral sites  $K_o$  were calculated by using the following relation;

$$K_t = 7.62 \times M_1 \times v_1^2 \times 10^{-7} \text{ N/m}$$

$$K_o = 10.62 \times \frac{M_2}{2} \times v_2^2 \times 10^{-7} \text{ N/m}$$

where  $M_1$  is the molecular weight of cations at the tetrahedral site,  $M_2$  is the molecular weight of cations at the octahedral site,  $v_1$  is the frequency band at tetrahedral site and  $v_2$  is also the frequency band at the octahedral site. The elastic force constants for tetrahedral ( $K_t$ ) and octahedral site ( $K_o$ ) for  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material are found to be  $1.5 \times 10^5$  dyne/cm and  $1.1 \times 10^5$  dyne/cm. This confirms that the elastic force constant of  $K_t$  is larger than that of  $K_o$ , which is associated with the shorter bond length of the tetrahedral cluster and longer bond length of octahedral cluster in  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material. It knows that bond length and the force constants have an inverse proportionality [17].

### 3.3. Elastic properties

The elastic properties study of solid materials are important for investigating the strength of binding forces in materials. Elastic properties of the ferrite materials can be estimated from the obtained XRD and FT-IR analysis. The stiffness constants  $C_{11}$  and  $C_{12}$  of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material were calculated using the relations [16] :

$$C_{11} = \frac{K_T + K_O}{2a}$$

$$C_{12} = \frac{C_{11}}{(1-\sigma)}$$

where 'a' is lattice constant and  $\sigma$  is the poisons ratio. The elastic moduli of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material were also calculated using the following relations [18,19]:

$$B = \frac{1}{3}(C_{11} + 2C_{12})$$

$$Y = \frac{(C_{11} - C_{12}) + (C_{11} + 2C_{12})}{C_{11} + C_{12}}$$

$$G = \frac{Y}{2(1+\sigma)}$$

where B is bulk modulus, Y is Young's modulus and G is rigidity modulus. The longitudinal wave velocity  $V_L$ , the transverse wave velocity  $V_s$  and the mean velocity  $V_m$  of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material were also evaluated using the relations [18,20];

$$V_L = \left(\frac{C_{11}}{D_b}\right)^{1/2}$$

$$V_s = \frac{V_L}{\sqrt{3}}$$

$$V_m = \left[\frac{1}{3}\left(\frac{2}{V_s^3} + \frac{1}{V_L^3}\right)\right]^{-1/3}$$

where  $D_b$  is the bulk density. The poisons ratio  $\sigma$  of the material was also calculated using the relation [20];

$$\sigma = \frac{v_L^2 - 2v_s^2}{2(v_L^2 - v_s^2)}$$

The Debye temperature  $\theta_D$  is a fundamental attribute of solid connecting elastic properties with thermodynamic properties such as specific heat. The Debye temperature of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material was calculated by using the relation [18];

$$\theta_D = \frac{hV_m}{k} \left[\left(\frac{3\rho q N_A}{4\pi M}\right)\right]^{1/3}$$

where h is Planck constant ( $6.626 \times 10^{-34}$  Js), k is Boltzmann constant ( $1.38 \times 10^{-23}$  J/K),  $N_A$  is Avogadro's number ( $6.022 \times 10^{23}$  1/mol), M is molecular weight of the sample, q is number of atoms in the unit formula,  $\rho$  is the density of the sample. The values of  $C_{11}$  and  $C_{12}$  are found to be 156 GPa and 53.4 GPa, respectively. This indicates that  $C_{11}$  is significantly larger than that of  $C_{12}$ . It is also observed that the calculated  $C_{11}$  and  $C_{12}$  are positive and satisfied the mechanical stability condition in a cubic crystal, which is  $(C_{11} - C_{12}) > 0$ ,  $(C_{11} + 2C_{12}) > 0$ ,  $C_{11} > 0$  and  $C_{12} < B < C_{11}$ . This suggests that both the synthesized  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material is mechanically stable [21].

The calculated values of the bulk modulus (B), Young's modulus (Y), and rigidity modulus (G) of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material are found to be 87.6, 1.75, and 69.7 GPa, respectively. The longitudinal wave velocity ( $V_L$ ) of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material is also found to be 6261 m/s, which is higher than the value of the shear wave velocity ( $V_s$ ), 3614.8 m/s. According to a previous report [16], when a wave travels through a material, it makes the particles vibrate. The vibrating particles collide with neighboring particles, which results in other particles to vibrate and longitudinal and shear elastic waves are produced. The particles in a material vibrates perpendicular to the direction of propagation of wave motion during the shear wave, and hence it requires a larger energy to make the neighboring particle vibrate. This, in turn, results in decrease in energy of wave velocity, and hence the shear wave velocity is nearly one-half of its longitudinal wave velocity [16].

The value of Poisson's ratio is found to be 0.24 for  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material. The obtained values lie in the range between -1 and 0.5, which is in conformity with theory of isotropic elasticity. The Debye temperature is an important parameter, which helps to understand the variations in thermodynamic properties such as the maximum atomic displacement, specific heat, melting temperature, or vibrational entropy [16]. It is the temperature at which the maximum lattice vibrations take place. In this study, the Debye temperature  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material is found to be 384.3 K.

#### 4. CONCLUSIONS

$\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material was successfully synthesized by sol-gel combustion method using citric acid as a fuel and chelating agent. The XRD study confirmed the formation of a cubic spinel-type  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material with lattice parameters and an average crystallite size of 0.8344 nm and 61.6 nm, respectively. The room temperature FTIR spectroscopic study of  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material confirmed the formation of two strong absorption bands. The higher frequency band which appeared at  $587\text{ cm}^{-1}$  assigned to the stretching vibration of the metal-oxygen bonding force of tetrahedral. The lower frequency band that appeared at  $426.5\text{ cm}^{-1}$  represented the metal-oxygen stretching vibration of octahedral group. The combined results of XRD and FTIR confirmed that the substitution of  $\text{Co}^{2+}$  and  $\text{Mn}^{2+}$  for  $\text{Ni}^{2+}$  did not change the basic structure of the spinel  $\text{NiFe}_2\text{O}_4$  material. The calculated force constants of the tetrahedral site  $K_T$  and octahedral site  $K_o$ , elastic constants, electric moduli, elastic velocities and Debye temperature were also matched with the reported values for  $\text{NiFe}_2\text{O}_4$  material. From the elastic property study, it was identified that the synthesized  $\text{Ni}_{0.8}\text{Co}_{0.15}\text{Mn}_{0.05}\text{Fe}_2\text{O}_4$  material is mechanically stable.

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